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RESEARCH ARTICLE

## Heterophase synthesis of cobalt ferrite

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### Abstract

**Objectives.** The study aimed to develop new methods for the synthesis of cobalt ferrite ( $\text{CoFe}_2\text{O}_4$ ), which is a precursor for the synthesis of  $\text{CoFe}_2\text{O}_4$ -based functional materials, as well as to study the physicochemical properties of the obtained phases.

**Methods.** Two methods were used for the synthesis of  $\text{CoFe}_2\text{O}_4$ : (1) heterophase interaction of hydrated iron oxide with cobalt(II, III) oxide and (2) heterophase interaction of hydrated iron oxide with an aqueous solution of cobalt(II) sulfate ( $C_{\text{Co}} = 0.147 \text{ mol/L}$ , solid/liquid = 1:43). In both cases, the precursor was hydrated iron oxide ( $\text{Fe}_2\text{O}_3$ , 84.4 wt %), which was obtained by the heterophase interaction of iron(III) chloride with a concentrated ammonia solution (6.0–9.5 mol/L). The resulting intermediate products were subjected to thermal treatment at 750°C (synthesis **1**) and at 900°C (synthesis **2**) for 10–30 h in increments of 10 h. The synthesized phases and products of their thermolysis were studied by differential thermal analysis and differential thermogravimetry (DTA–DTG), X-ray diffraction analysis (XRDA), and granulometry.

**Results.** The hydrated iron oxide sample remained amorphous even up to the crystallization temperature of 445°C, which corresponds to the exothermic effect on the DTA curve. Further heating led to the  $\alpha$ -modification of iron(III) oxide of the hexagonal system ( $a = b = 5.037 \pm 0.002 \text{ Å}$ ;  $c = 13.74 \pm 0.01 \text{ Å}$ ), which has an average particle size of 1.1  $\mu\text{m}$ . XRDA results showed that a synthesis temperature of 750°C (synthesis **1**) and a heat treatment duration of 30 h were sufficient for the formation of a single-phase cobalt ferrite ( $a = 8.388 \pm 0.002 \text{ Å}$ ) with an average particle diameter of 1.9  $\mu\text{m}$ . For synthesis **2**, a higher temperature of 900°C was

used because sample weight loss (about 12.5%) was observed in the temperature range of 720–810 °C based on the DTA results, which was due to the removal of  $\text{SO}_2$  and  $\text{SO}_3$ . Moreover, when synthesis temperature and duration were at 900 °C and 30 h, respectively,  $\text{CoFe}_2\text{O}_4$  with  $a = 8.389 \pm 0.002 \text{ \AA}$  was formed. The results of the granulometric analysis showed that particles of different diameters were formed. The smallest particle size (1.5  $\mu\text{m}$ ) of cobalt ferrite was obtained by the heterophase interaction of hydrated iron(III) oxide ( $\text{Fe}_2\text{O}_3$ , 84.4 wt %) with an aqueous solution of cobalt sulfate with  $C_{\text{Co}} = 0.147 \text{ mol/L}$ .

**Conclusions.** Depending on the method used for the synthesis of cobalt ferrite, particles of different diameters are formed. The smallest particle size of cobalt ferrite was obtained from the heterophase interaction of hydrated iron(III) oxide with an aqueous solution of cobalt(II) sulfate.

**Keywords:** iron, cobalt, ferrite, oxides, thermal analysis, X-ray phase analysis, particle size analysis

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## НАУЧНАЯ СТАТЬЯ

# Гетерофазный синтез феррита кобальта

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### Аннотация

**Цели.** Разработка новых методов синтеза феррита кобальта ( $\text{CoFe}_2\text{O}_4$ ), являющегося предшественником для синтеза функциональных материалов на его основе, а также исследование физико-химических свойств полученных фаз.

**Методы.** Гидратированный оксид железа и феррит кобальта получали гетерофазным методом. Синтезированные фазы и продукты их термолитиза изучали методами дифференциально-термического анализа и дифференциальной термогравиметрии (ДТА–ДТГ), рентгенофазового анализа (РФА) и гранулометрии.

**Результаты.** В статье изложены результаты двух методов синтеза феррита кобальта ( $\text{CoFe}_2\text{O}_4$ ) и исследования полученных фаз. В обоих случаях в качестве предшественника выступал гидратированный оксид железа(III) с содержанием  $\text{Fe}_2\text{O}_3$  – 84.4 мас. %, полученный гетерофазным взаимодействием хлорида железа(III) с концентрированным раствором аммиака (6.0–9.5 моль/л). Первый способ заключался во взаимодействии гидратированного оксида железа(III) с оксидом кобальта(II, III), второй – во взаимодействии гидратированного оксида железа(III) с водным раствором сульфата кобальта(II) с концентрацией  $C_{\text{Co}} = 0.147 \text{ моль/л}$  ( $T : Ж = 1 : 43$ ). Получившиеся промежуточные продукты подвергали термической обработке при 750 °C (синтез **1**) и 900 °C (синтез **2**) в течение 10–30 ч с шагом 10 ч.

**Выводы.** Феррит кобальта ( $\text{CoFe}_2\text{O}_4$ ) получен двумя способами. С использованием комплекса методов (РФА, ДТА–ДТГ, гранулометрии) исследованы физико-химические свойства синтезированных образцов. Установлено, что гидратированный оксид железа(III) вплоть до температуры кристаллизации ( $445^\circ\text{C}$ ), соответствующей экзотермическому эффекту на кривой ДТА, остается рентгеноаморфным. Дальнейшее нагревание его приводит к образованию  $\alpha$ -модификации оксида железа(III) гексагональной сингонии ( $a = b = 5.037 \pm 0.002 \text{ \AA}$ ;  $c = 13.74 \pm 0.01 \text{ \AA}$ ), средний размер частиц которой равен  $1.1 \text{ мкм}$ . Согласно данным РФА, в синтезе **1** при  $750^\circ\text{C}$  и продолжительности термообработки  $30 \text{ ч}$  образуется однофазный феррит кобальта ( $a = 8.388 \pm 0.002 \text{ \AA}$ ) со средним диаметром частиц  $1.9 \text{ мкм}$ . В интервале температур  $720\text{--}810^\circ\text{C}$  в образце наблюдается убыль массы (около  $12.5\%$ ), связанная с удалением  $\text{SO}_2$  и  $\text{SO}_3$ . Поэтому в синтезе **2** температуру нагревания увеличивали до  $900^\circ\text{C}$ . Показано, что при  $900^\circ\text{C}$  и продолжительности синтеза  $30 \text{ ч}$  также образуется феррит кобальта ( $\text{CoFe}_2\text{O}_4$ ) ( $a = 8.389 \pm 0.002 \text{ \AA}$ ). Результаты гранулометрического анализа указывают на зависимость диаметра образующихся частиц от способа получения феррита кобальта. Наименьший размер частиц ( $1.5 \text{ мкм}$ ) обнаружен у феррита кобальта, полученного гетерофазным взаимодействием гидратированного оксида железа(III) ( $\text{Fe}_2\text{O}_3$  –  $84.4 \text{ мас. \%}$ ) с водным раствором сульфата кобальта с концентрацией  $C(\text{Co}^{2+}) = 0.147 \text{ моль/л}$ .

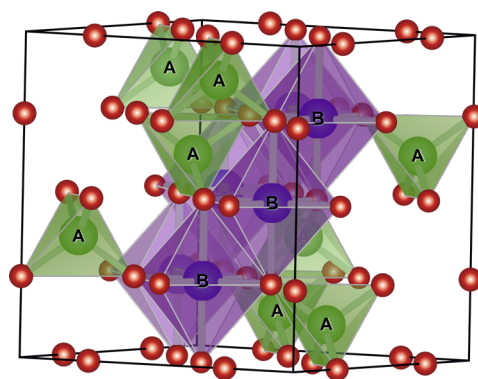
**Ключевые слова:** железо, кобальт, феррит, оксиды, термический анализ, рентгенофазовый анализ, гранулометрический анализ

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## INTRODUCTION

Due to their unique properties, spinel-structured ferrites are widely used in various fields of science and technology. This class of magnetic ceramics is used in various applications, such as information storage systems, magnetic fluids, gas sensors, catalysts, rechargeable lithium batteries, magnetic cores, microwave absorbers, medical diagnostics and therapy, wastewater treatment, and biosensors [1–7].

The ferrite formula can be represented as  $\text{AB}_2\text{O}_4$ . It has a spinel structure (Fig. 1), where the tetrahedral A-position is occupied by ions in the oxidation state of +2 (e.g.,  $\text{Mg}^{2+}$ ,  $\text{Fe}^{2+}$ ,  $\text{Ni}^{2+}$ ,  $\text{Co}^{2+}$ , and  $\text{Mn}^{2+}$ ). The octahedral B-position is mainly occupied by  $\text{Fe}^{3+}$  ions, but they can be replaced by other ions in the oxidation state of +3 (e.g.,  $\text{Al}^{3+}$ ,  $\text{Cr}^{3+}$ ). If the A-position is occupied by ions in the +3 oxidation state and the B-position is equally populated by ions in the +2 and +3 oxidation states, the spinel structure is called reversed [8, 9].



**Fig. 1.** Ferrite spinel structure [9].

One of the representatives of this class of materials is cobalt ferrite ( $\text{CoFe}_2\text{O}_4$ ), a well-known magnetic hard material with a high coercive force associated with a small size of crystalline grains and strong magnetocrystalline

anisotropy and moderate magnetization. It has an inverted spinel structure, where all or most of the  $\text{Co}^{2+}$  ions occupy an octahedral B-position and  $\text{Fe}^{3+}$  ions occupy both tetrahedral (A) and octahedral (B) positions.

These properties, along with high chemical stability, make it possible to use cobalt ferrite in various applications, such as medicine, catalysis, magnetic data carriers in audio and video cassettes, high-density digital recording discs, various sensors, recording devices, and solar cells [10–13]. The purpose of this work was to develop methods for the synthesis of cobalt ferrite, which is a precursor for the synthesis of  $\text{CoFe}_2\text{O}_4$ -based functional materials, as well as to study the physicochemical properties of the obtained phases.

## MATERIALS AND METHODS

Iron(III) chloride hexahydrate, which was used as the starting material in the study, cobalt(II, III) oxide, and cobalt sulfate were supplied by *Merck*, Germany, while aqueous ammonia was sourced from *Himmed*, Russia.

Hydrated iron(III) oxide was obtained by the heterophase interaction of iron(III) chloride with a concentrated solution of ammonia hydrate (6.0–9.5 mol/L).

Cobalt ferrite was synthesized using two heterophase synthesis methods.

**Synthesis 1.** Samples of hydrated iron(III) oxide and cobalt(II,III) oxide in the ratio of 1.5:1 were placed in a glass flask with a lapped stopper. The hydrated iron oxide to cobalt oxide ratio used was based on the oxide content of hydrated iron oxide and  $\text{CoFe}_2\text{O}_4$ . Distilled water was poured into the flask with the substance, and the flask containing the reaction mixture was shaken on a vibrating apparatus at room temperature ( $22 \pm 2^\circ\text{C}$ ) for 10 h. The resulting suspension was transferred to a crucible and subjected to slow evaporation. Then, the intermediate product obtained was subjected to heat treatment at  $750^\circ\text{C}$  for 10–30 h in increments of 10 h.

**Synthesis 2.** A suspension of hydrated iron(III) oxide was placed in a glass flask with a lapped stopper, to which a solution of cobalt(II) sulfate with  $C_{\text{Co}} = 0.147$  mol/L (solid/liquid = 1:43) was added. Then, the flask was shaken on a vibrating apparatus at room temperature for 10 h. The resulting suspension was transferred to a crucible and subjected to slow evaporation, and the intermediate product obtained was subjected to heat treatment at  $900^\circ\text{C}$  for 30 h in increments of 10 h.

The oxide content in hydrated iron(III) oxide was determined by gravimetric analysis.

Differential thermal analysis and differential thermogravimetry (DTA–DTG) of the samples was conducted on a Q-1500 D air derivatograph (*MOM*, Hungary) with simultaneous recording of four curves: differential (DTA), temperature ( $T$ ), differential thermogravimetric (DTG), and integrated mass change (TG) curves using a hardware–software complex in LabVIEW 8.21 (*National Instruments*, USA). The temperature was measured with a platinum–platinum rhodium thermocouple (PP-1) with an error of  $\pm 5^\circ\text{C}$  in the temperature range of 20– $1000^\circ\text{C}$ , using  $\alpha\text{-Al}_2\text{O}_3$  as a standard.

X-ray diffraction analysis (XRDA) was carried out on a D8 Advance diffractometer (*Bruker*, Germany) under  $\text{SiK}_\alpha$  radiation with continuous rotation of the sample ( $1^\circ/\text{min}$ , step-by-step  $2\theta = 0.02^\circ$ , exposure 0.5 s) and modes in the angle range  $2\theta = 5^\circ\text{--}80^\circ$ . An ICDD card file was used for phase identification.

Granulometric analysis (determination of the particle size distribution function) was performed using a DelsaNano laser particle analyzer (*Beckman Coulter*, USA). Specific surface area and pore size were measured using a SA 3100 analyzer (*Beckman Coulter*, USA). Surface area was calculated from adsorption isotherms using the Brunauer–Emmett–Teller (BET) method. A nonporous sample ASX\_1\_4 (specific surface area according to BET was  $4.18\text{ m}^2/\text{g}$ ) GSO 9735\_2010<sup>1</sup> (a set of standard samples of nanostructured aluminum oxide ASX\_1) was used as a standard. The measurement range of the specific surface was  $0.1\text{--}2000\text{ m}^2/\text{g}$ . The resolution of the pressure sensor was 0.0062 kPa. The limit of the permissible relative error of the analyzer when measuring the specific surface was  $\pm 5\%$ . Preliminary degassing of the samples was carried out for 1 h at  $t = 70^\circ\text{C}$  and a pressure of 0.1 Pa.

Heat treatment of the samples was conducted in a TK-12.1250.N.1F laboratory furnace with a Thermomatic-N automatic temperature controller that has an error of  $\pm 1^\circ\text{C}$  at nominal temperature.

## RESULTS AND DISCUSSION

The heterophase synthesis of hydrated iron(III) oxide involves the interaction of solid ferric chloride with a concentrated ammonia solution (6.0–9.5 mol/L). The synthesis proceeded at a high rate at room temperature. Simultaneously, a well-filtered hydrated iron(III) oxide powder with a high content of  $\text{Fe}_2\text{O}_3$  (84.4 wt %) was obtained, which was easily

<sup>1</sup> [https://gssso.ru/wp-content/uploads/2016/08/catalog\\_gso\\_2020\\_3.pdf](https://gssso.ru/wp-content/uploads/2016/08/catalog_gso_2020_3.pdf)

washed off from impurities and retained a high reactivity. When obtaining hydrated iron(III) oxide, the prewashing operation is important, and it was conducted by the repeated treatment of the sediment with distilled water. Washing removes chloride ions in hydrated iron(III) oxide precipitate, thereby lowering its content to less than 0.05 wt %. Figure 2 shows the particle size distribution of hydrated iron(III) oxide. On the histogram (Fig. 2), there is a pronounced maximum corresponding to the maximum amount of powder particles with a size of 0.8–1.2  $\mu\text{m}$ . Hence, it can be assumed that the hydrated iron(III) oxide was a relatively monodisperse powder. About 95% was accounted for by particles with a size of 0.7–2.0  $\mu\text{m}$ .

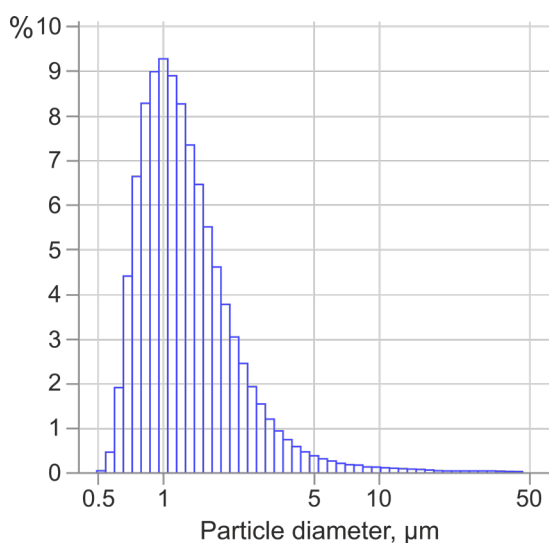


Fig. 2. Particle size distribution of hydrated iron(III) oxide.

Studies have shown that the average particle size is 1.1  $\mu\text{m}$ . The specific surface area of hydrated iron(III) oxide measured by the BET method was 76.72  $\text{m}^2/\text{g}$ .

Thermolysis of hydrated iron(III) oxide was studied in the temperature range of 20–1000°C. Figure 3 shows the results of the DTA. It was found that the loss of the bulk of water in hydrated iron(III) oxide at temperatures as high as 200°C corresponds to the endothermic effect on the DTA curve (Fig. 3). The resulting phase remains amorphous, which was confirmed by the results of XRDA. With a further increase in temperature, there are no other endoeffects that could indicate a stepwise dehydration. The exoeffect at 445°C corresponds to the transition from an amorphous state to a crystalline state. Further heating above the specified temperature is accompanied by the formation of crystalline oxide  $\text{Fe}_2\text{O}_3$ . Figure 4 shows the diffractogram of the thermolysis products of hydrated iron(III) oxide at 500°C. Obviously, at 500°C, a hexagonal  $\alpha$ -modification of iron(III) oxide was formed. The parameters of the  $\alpha$ - $\text{Fe}_2\text{O}_3$  lattice are as follows:  $a = b = 5.037 \pm 0.002 \text{ \AA}$  and  $c = 13.74 \pm 0.01 \text{ \AA}$ .

To establish the effect of the precursor on the phase composition of cobalt ferrite, two synthesis methods were conducted with hydrated iron(III) oxide acting as a precursor in both. Either cobalt oxide  $\text{Co}_3\text{O}_4$  (synthesis 1) was added to it, or an aqueous solution of cobalt(II) sulfate ( $C_{\text{Co}} = 0.147 \text{ mol/L}$ , synthesis 2).

Figures 5 and 6 show the results of the thermal analysis of the intermediate products obtained during syntheses after evaporation of suspensions.

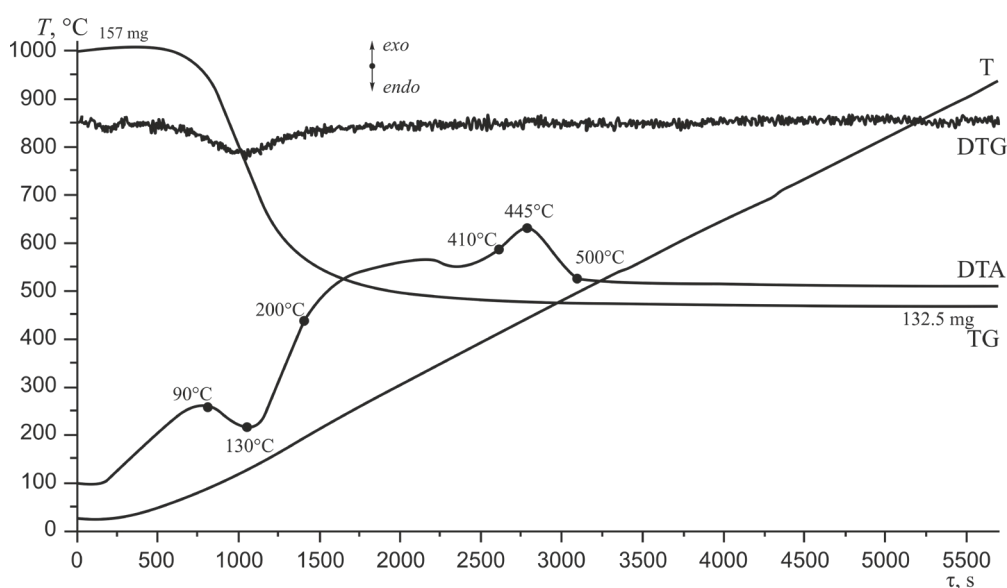


Fig. 3. Thermogravigram of hydrated iron(III) oxide ( $\text{Fe}_2\text{O}_3$ , 84.4 wt %).



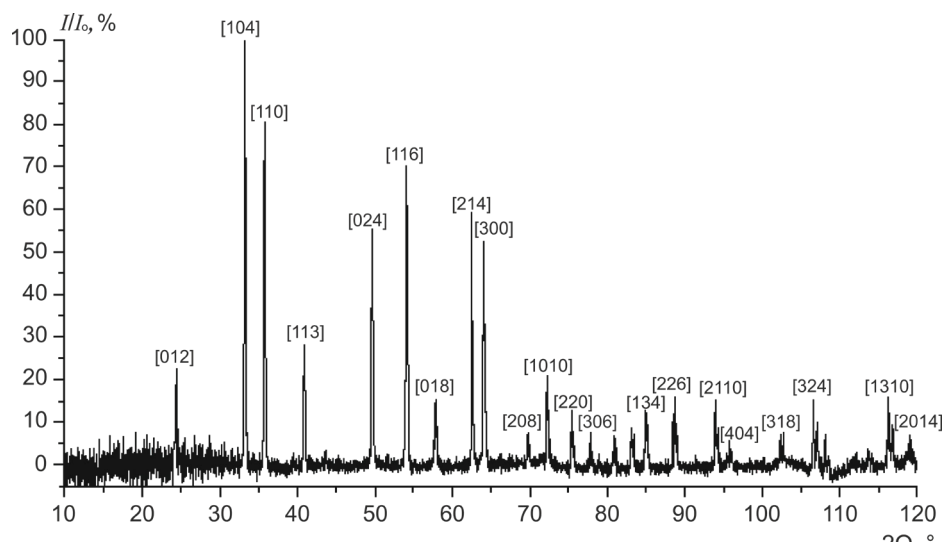


Fig. 4. X-ray diffraction pattern of the thermal decomposition product of hydrated iron(III) oxide,  $T = 500^{\circ}\text{C}$ .

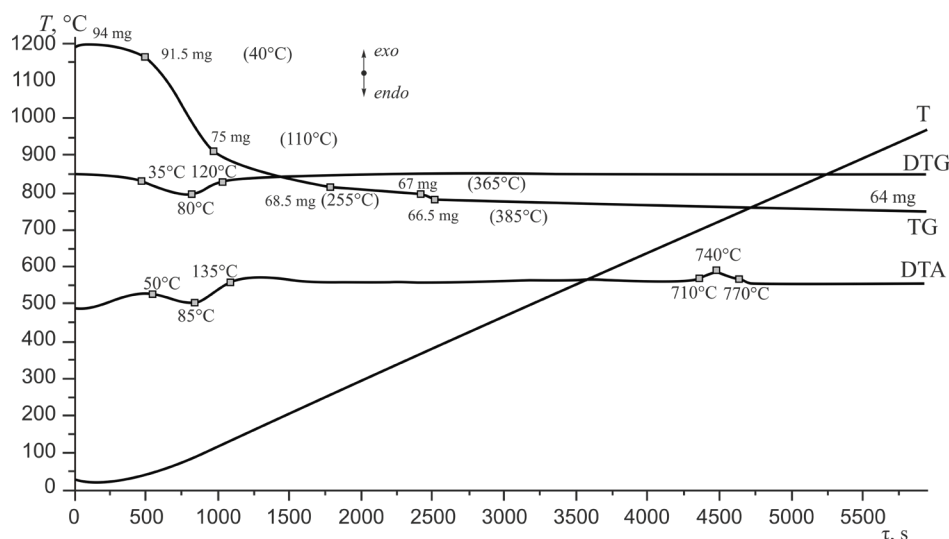


Fig. 5. Thermogravigram of the intermediate product obtained by the interaction of hydrated iron(III) oxide with cobalt(II, III) oxide (synthesis 1).

Thermogravigrams showed an endothermic effect associated with the dehydration processes ( $50\text{--}135^{\circ}\text{C}$ , Fig. 5;  $75\text{--}160^{\circ}\text{C}$  and  $275\text{--}330^{\circ}\text{C}$ , Fig. 6) and with the removal of  $\text{SO}_2$  and  $\text{SO}_3$  ( $725\text{--}805^{\circ}\text{C}$  and  $805\text{--}870^{\circ}\text{C}$  [15], Fig. 6). In both cases, there was no exothermic effect at  $445^{\circ}\text{C}$  that corresponds to the crystallization of individual hydrated iron(III) oxide. On the thermogravigram of the intermediate product obtained by the interaction of hydrated iron(III) oxide with cobalt(II, III) oxide (Fig. 5), an exoeffect was observed at  $710\text{--}770^{\circ}\text{C}$ , which can be associated with the formation of a compound of a given composition, but the sample annealed at  $710^{\circ}\text{C}$  was amorphous.

The resulting intermediates were subjected to heat treatment. In the case of synthesis 1, a temperature of  $750^{\circ}\text{C}$  was used, corresponding to the

exothermic effect on the DTA curve (Fig. 5), and the synthesis duration was 10–30 h in increments of 10 h. In all cases, fine crystalline powder of  $\text{CoFe}_2\text{O}_4$  (cubic syngony) was obtained, which is confirmed by the results of XRDA (ICDD 79-1744:  $a = 8.390 \text{ \AA}$ ). However, in the case of 10-h annealing, the diffraction peaks of unreacted iron and cobalt oxides were observed on the diffractogram (Fig. 7). Annealing for 30 h turned out to be the most optimal—pronounced peaks of only one phase are present on the diffractogram, which is cubic ( $\text{CoFe}_2\text{O}_4$ , Fig. 8). The calculated lattice parameter of the synthesized cobalt ferrite is  $a = 8.388 \pm 0.002 \text{ \AA}$ .

In synthesis 2, an annealing temperature of  $900^{\circ}\text{C}$  was selected. This is due to the fact that up to a temperature of  $870^{\circ}\text{C}$ , there was a decrease in

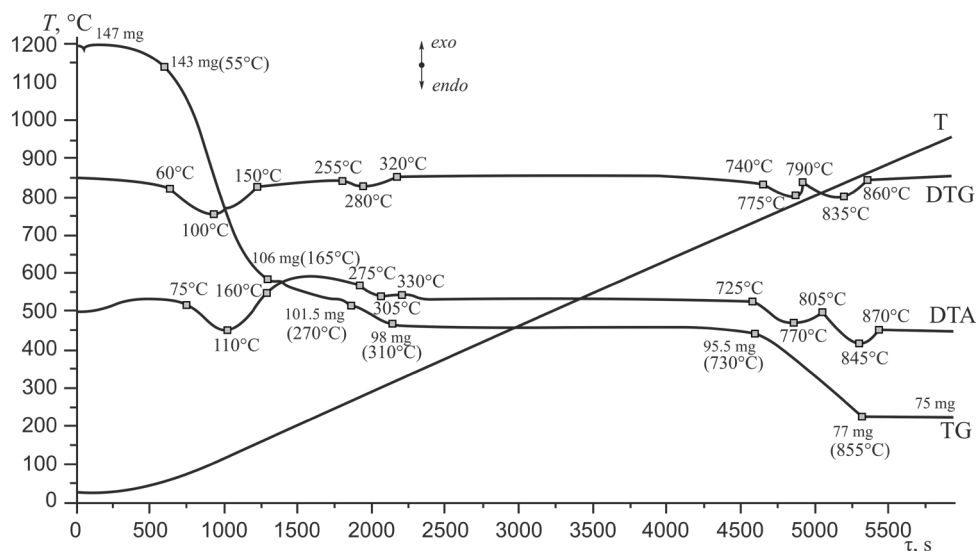


Fig. 6. Thermogravimogram of the intermediate product obtained by the interaction of hydrated iron(III) oxide with a solution of cobalt(II) sulfate (synthesis 2).

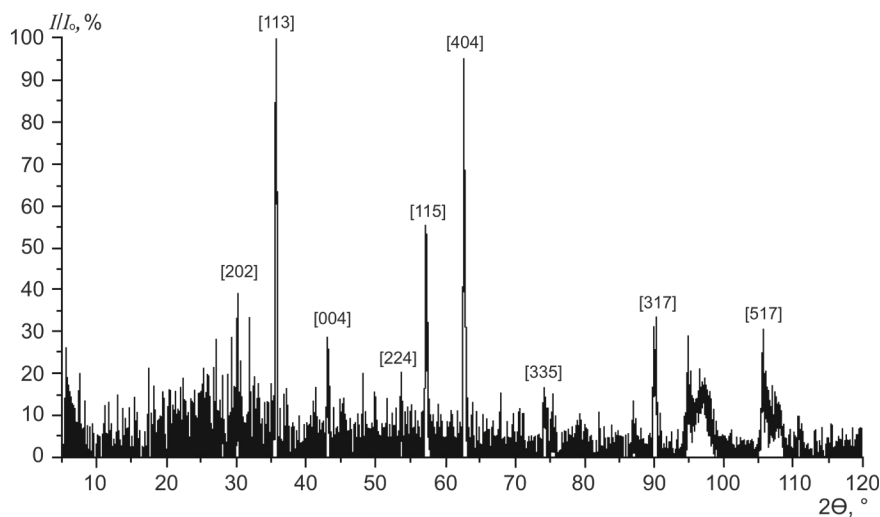


Fig. 7. X-ray diffraction pattern of cobalt ferrite synthesized at 750°C and annealed for 10 h (synthesis 1).

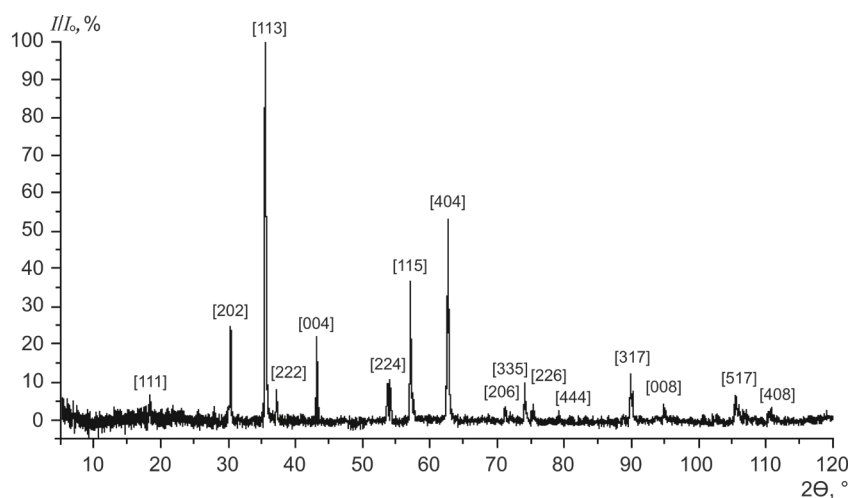
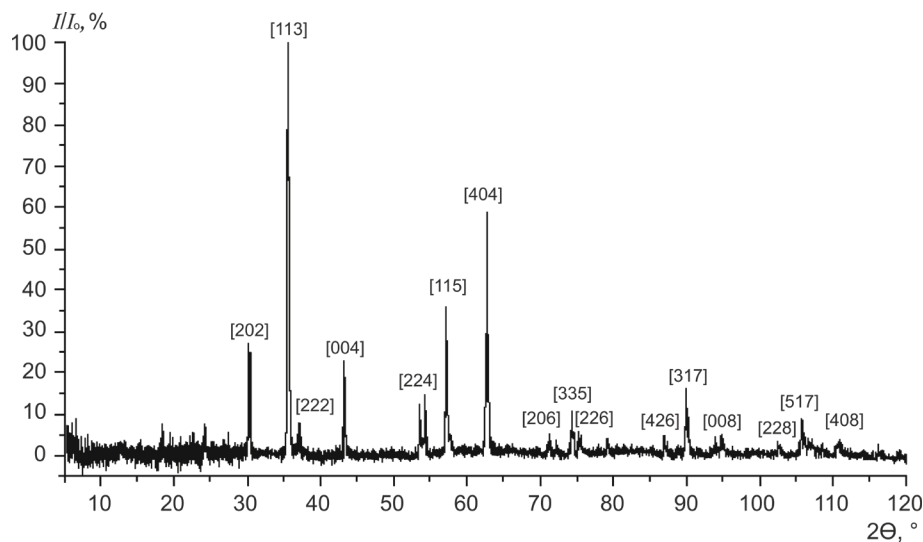
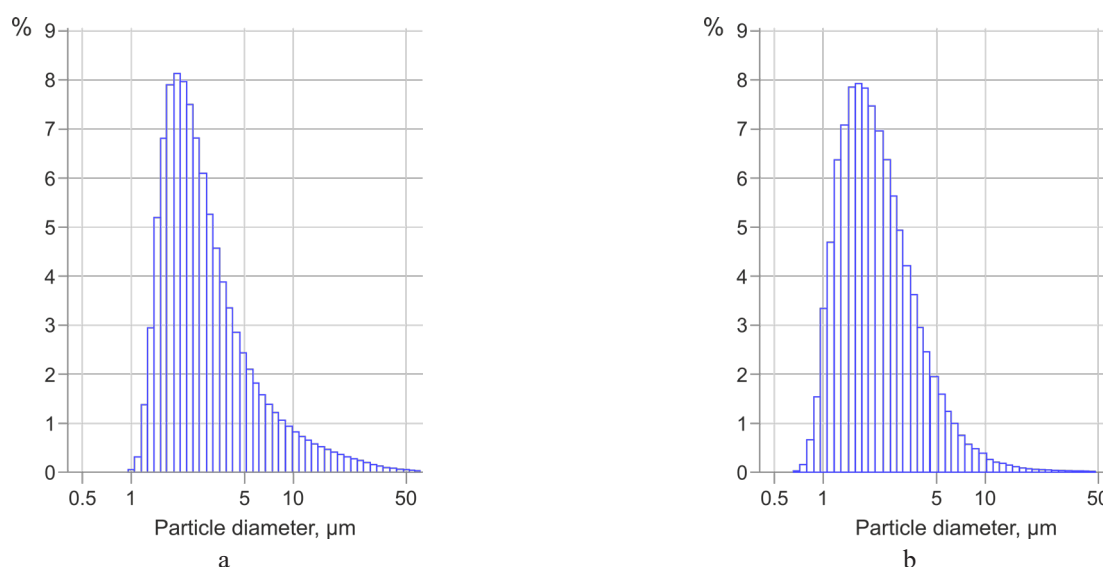


Fig. 8. X-ray diffraction pattern of cobalt ferrite synthesized at 750°C and annealed for 30 h (synthesis 1).



**Fig. 9.** X-ray diffraction pattern of cobalt ferrite synthesized at 900°C and annealed for 30 h (synthesis 2).



**Fig. 10.** Particle size distribution of cobalt ferrite: (a) synthesis 1, 750°C, 30 h; (b) synthesis 2, 900°C, 30 h.

sample mass (about 12.5%, Fig. 6). The duration of annealing was 30 h. Figure 9 shows the diffractogram of the resulting product. The lattice parameter of the synthesized ferrite is  $a = 8.389 \pm 0.002$  Å.

The influence of the choice of precursors on the size of cobalt ferrite particles has been studied. Figure 10a shows the size distribution of cobalt ferrite particles (synthesis 1, 750°C, 30 h). On the histogram (Fig. 10a), there is a pronounced maximum corresponding to the maximum presence of powder particles with a size of 1.5–3.0 μm. The main fraction consisted of particles with a size of 1.0–5.0 μm (~85%) with an average diameter of 1.9 μm (for the initial hydrated iron(III) oxide, it was 1.1 μm).

Figure 10b shows the size distribution of cobalt ferrite particles (synthesis 2, 900°C, 30 h). In this case, it is obvious from the histogram that a maximum

number of cobalt ferrite powder particles have a size of 1.2–3.0 μm, and the average particle diameter was 1.5 μm.

As can be seen from the distributions, the smallest particles were formed in cobalt ferrite obtained by the heterophase interaction of hydrated iron(III) oxide ( $\text{Fe}_2\text{O}_3$ , 84.4 wt %) with an aqueous solution of cobalt(II) sulfate with  $C_{\text{Co}} = 0.147$  mol/L.

## CONCLUSIONS

$\text{CoFe}_2\text{O}_4$  was synthesized in two ways. The first method is the heterophase interaction of hydrated iron(III) oxide with cobalt(II, III) oxide with subsequent thermal treatment at 750°C for 30 h (synthesis 1), and the second method is the heterophase interaction of hydrated iron(III) oxide



( $\text{Fe}_2\text{O}_3$ , 84.4 wt %) with an aqueous solution of cobalt(II) sulfate ( $C_{\text{Co}} = 0.147 \text{ mol/L}$ ) with subsequent heat treatment at  $900^\circ\text{C}$  for 30 h (synthesis 2).

Using various methods (XRDA, DTA–DTG, and granulometry), the physicochemical properties of the synthesized samples were investigated. The results of the studies on hydrated iron(III) oxide showed that up to the crystallization temperature of  $445^\circ\text{C}$ , which corresponds to the exothermic effect on the DTA curve, the sample remained amorphous. Further heating led to the  $\alpha$ -modification of iron(III) oxide hexagonal syngony ( $a = b = 5.037 \pm 0.002 \text{ \AA}$ ;  $c = 13.74 \pm 0.01 \text{ \AA}$ ), which has an average particle size of  $1.1 \text{ \mu m}$ .

Using the XRD method, it was found that in the case of synthesis 1, a temperature of  $750^\circ\text{C}$  and a heat treatment for 30 h were sufficient to form a single-phase cobalt ferrite ( $a = 8.388 \pm 0.002 \text{ \AA}$ ) with an average particle diameter of  $1.9 \text{ \mu m}$ . The temperature used in synthesis 2 was increased to  $900^\circ\text{C}$  due to a decrease in sample mass (about 12.5%) observed during synthesis in the temperature range of  $720\text{--}810^\circ\text{C}$  based on the results of differential thermal analysis.

Loss in sample mass was due to the removal of  $\text{SO}_2$  and  $\text{SO}_3$ . Using a synthesis temperature and a synthesis duration of  $900^\circ\text{C}$  and 30 h, respectively,  $\text{CoFe}_2\text{O}_4$  with  $a = 8.389 \pm 0.002 \text{ \AA}$  was formed.

Depending on the synthesis method, cobalt ferrite particles of different diameters were formed. The smallest particle size of cobalt ferrite ( $1.5 \text{ \mu m}$ ) was obtained by the heterophase interaction of hydrated iron(III) oxide ( $\text{Fe}_2\text{O}_3$ , 84.4 wt %) with an aqueous solution of cobalt(II) sulfate with  $C_{\text{Co}} = 0.147 \text{ mol/L}$ .

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### Author's contribution

**Elena E. Nikishina** – conducting research, preparing a manuscript.

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